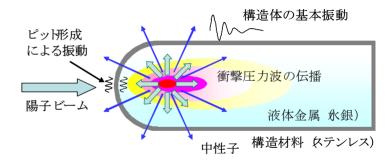
## Development of Advanced Materials for Spallation Neutron Sources and Radiation Damage Simulation Code Based on Multi-Scale Model\*

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- 1. High Energy Accelerator Research Organization (KEK),
- 2. Tohoku University,
- 3. Ibaraki University,
- 4. Hokkaido University,
- 5. Kyoto University and
- 6. Japan Atomic Energy Agency (JAEA)
- Super tungsten with high toughness and R.T. ductility
- >GBE stainless steel highly resistant to corrosion and radiation
- ➤ Multi-scale model simulation of radiation damage

# Objective



Development of materials for the pulsed spallation neutron sources with a MW-power injection of proton beam And investigation of damage process due to radiation and beam impact

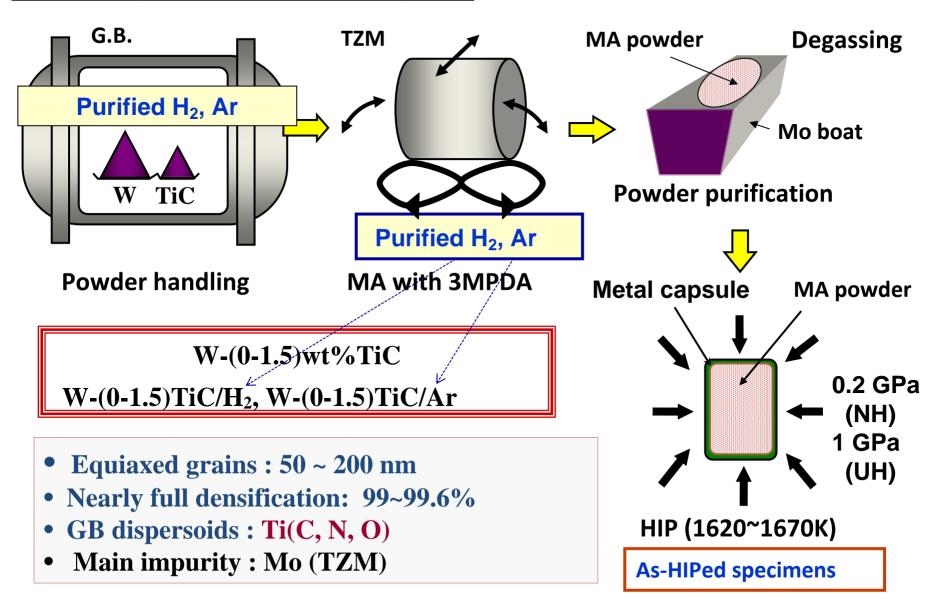
Material Needs for Long-life Source:

Highly tough tungsten for spallation NS and fusion reactor High corrosion- and radiation-resistance SUS (> 10 dpa)

Needs to Radiation Damage Analyses for ADS and pulsed source: Multi-scale model simulation

#### Super tungsten with high toughness and R.T. ductility (1)

Processing of UFG W-TiC compacts --- To be presented by H. Kurishita,



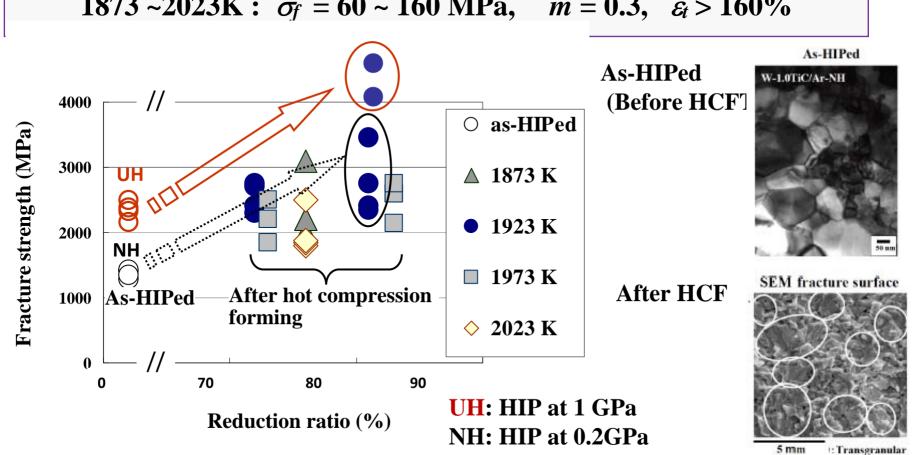
#### Super tungsten with high toughness and R.T. ductility (2)

#### Hot compression forming (HCF) in the recrystallized state

W- $(1\sim1.1)$ %TiC/Ar: 1GPa  $\rightarrow$  4.4GPa, RT ductility

#### **Condition**

1873 ~2023K:  $\sigma_f = 60 \sim 160 \text{ MPa}, \quad m = 0.3, \quad \varepsilon_t > 160\%$ 

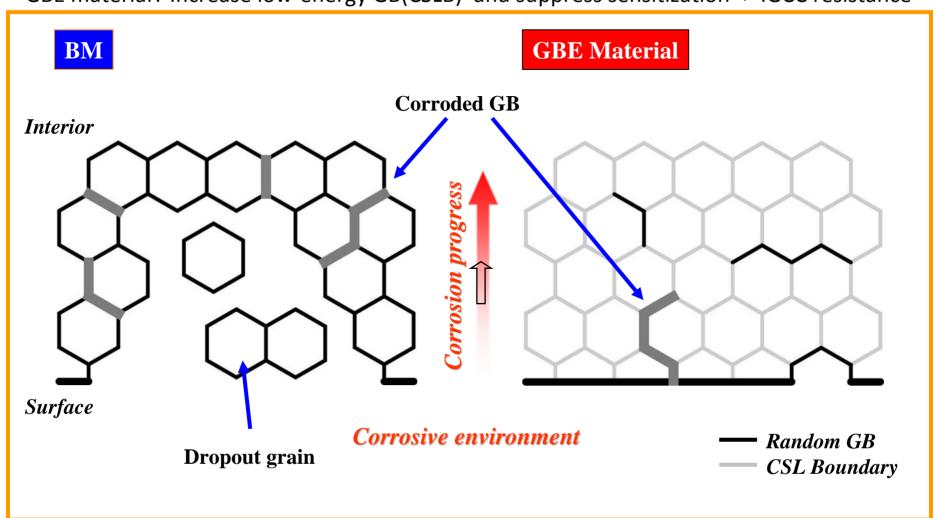


#### Recent activities on GBE materials (1)

#### Comparison of Corrosion Scheme between BM and GBE Material

BM:  $Cr_{23}C_6$  precipitation at random GB  $\rightarrow$  Depletion of Cr content  $\rightarrow$  Easily corroded

GBE material: Increase low-energy GB(CSLB) and suppress sensitization -> IGCC resistance



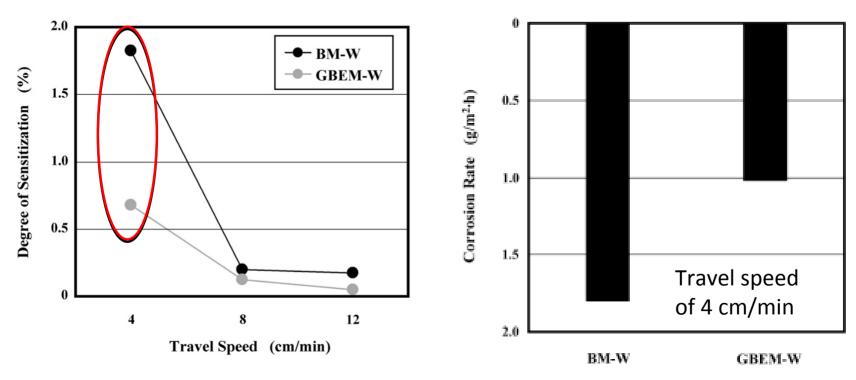
#### Solution Recent activities on GBE materials (2) Annealing Cold rolling Development of GBE 304SS **Heat Thermal process** OIM Map (粒界性格分布) **304 BM 304 GBEM Heat Process** Cold rolling → Annealing 5% 1200 K, 72 h - Random GB (high energy) — CSLB (low energy) 200 µm 200 µm CSL%: 63% **CSL%: 87% GBE 304 SS** Highly-dense CSLB Discontinuous random GB ( break) $(BM:63\% \rightarrow GBEM:87\%)$

Corrosion rate was reduced by a factor of 4

M. Shimada, H. Kokawa, Z.J. Wang, Y.S. Sato and I. Karibe, Acta Materialia, 50 (2002), 2331.

#### Recent activities on GBE materials (3)

Comparison of Degree of Sensitization (DOS) in the HAZ and the Corrosion Mass Loss between the Welded BM and GBEM.

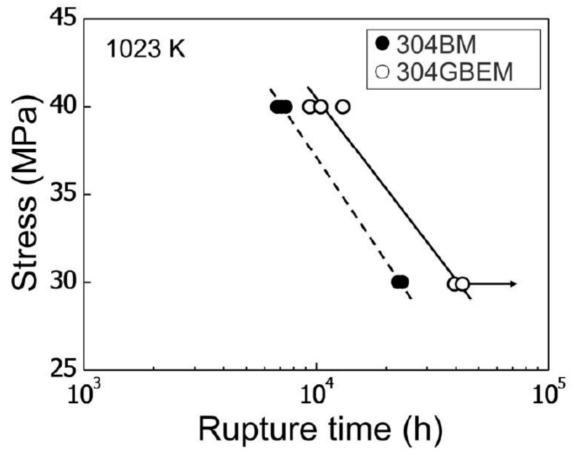


The corrosion rate of the weld-decay region in the GBEM-W was about half of that in the BM-W.

The corrosion rate in the GBEM-W should be much smaller than half of that in the BM-W, considering that the Streicher-tested specimen consisted of the whole weldment including other non-sensitized parts.

#### Recent activities on GBE materials (4)

#### Creep Test of SUS304 at 1023 K



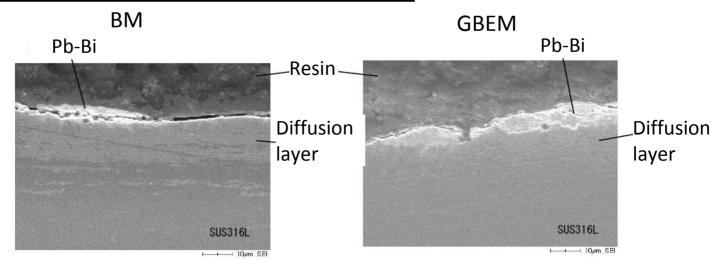
CSL has a possibility to suppress a grain boundary sliding. Rupture time of 304GBEM was 1.5 times larger than that of 304BM.

H. Kokawa, et al., Grain Boundary Engineering of Austenitic Stainless Steel, Bull. Of the iron and steel insitute of Japan, 15-12 (2010)

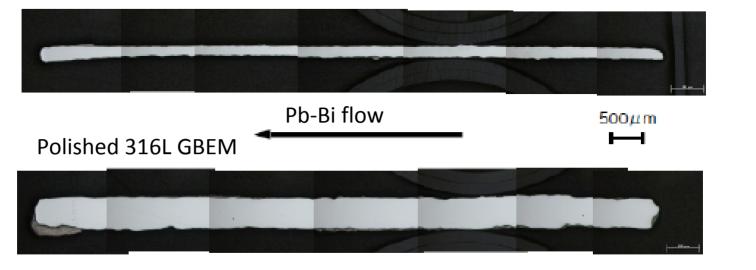
#### **Recent activities on GBE materials (5)**

--- Presented by S. Saito, 10/19 13:55 in IWSMT-10

#### Erosion test of BM and GBEM in Pb-Bi Flow



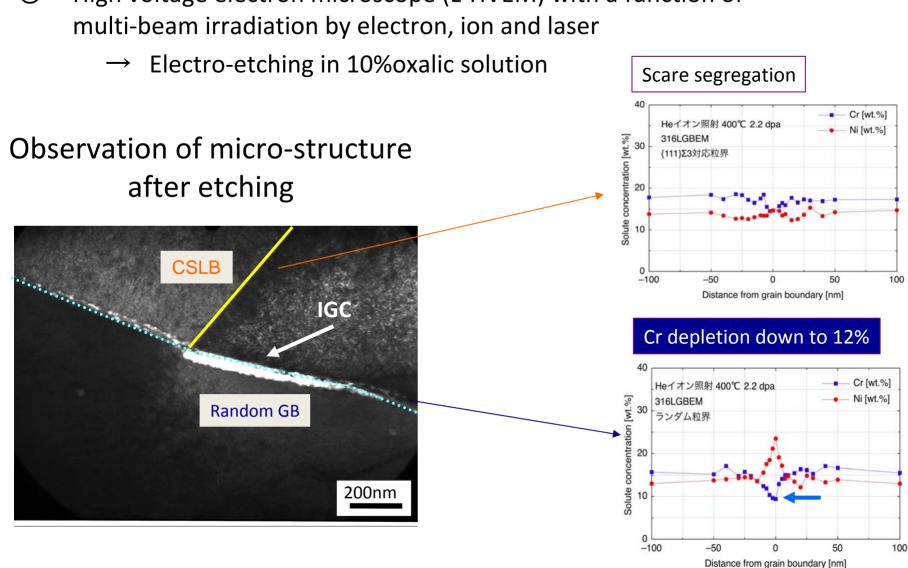
Polished 316L BM: remarkably eroded



#### Recent activities on GBE materials (6)

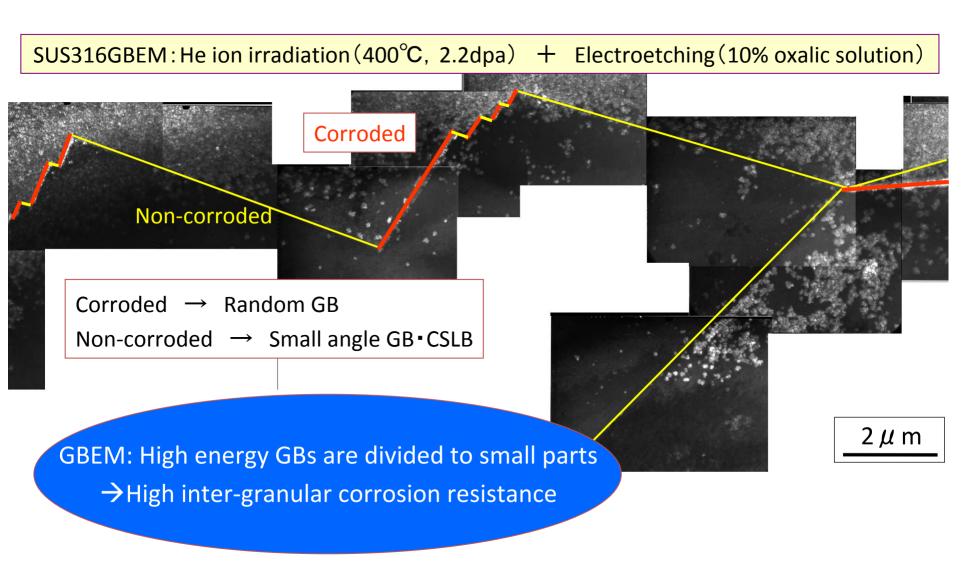
#### **Beam Irradiation Experiments**

0 High voltage electron microscope (L-HVEM) with a function of multi-beam irradiation by electron, ion and laser



#### Recent activities on GBE materials (8)

CLS's: Dividing into parts of IGC network



# MULTI-SCALE MODELING OF IRRADIATION EFFECTS IN SPALLATION NEUTRON SOURCE MATERIALS

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Yoshihisa Kaneko<sup>4</sup>, Masayoshi Kawai<sup>3</sup>, Ippei Kishida<sup>4</sup>, Satoshi Kunieda<sup>5</sup>, Koichi Sato<sup>1</sup>, Satoshi Shimakawa<sup>5</sup>, Futoshi Shimizu<sup>5</sup>, Satoshi Hashimoto<sup>4</sup>, Naoyuki Hashimoto<sup>6</sup>, Tokio Fukahori<sup>5</sup>, Yukinobu Watanabe<sup>7</sup>, Qiu Xu<sup>1</sup>, Shiori Ishino<sup>8</sup>

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<sup>7</sup> Kyushu University

<sup>8</sup>Univerity of Tokyo

- Nuclear reactions and electronic excitaion: PHITS, PKA energy spectrum
- Atomic collision : MD, k-Monte Carlo, Subcascade analysis.
- Damage structure evolution : Reaction kinetic analysis
- Mechanical properties changes : Discrete dislocation dynamics

#### **Multi-Scale Modeling (2) PHITS** Reaction kinetics analysis Nuclear reaction Dislocation dynamics k-Monte Carlo MD database High energy $10^{2}$ particle Irradiated Primary Knock-on Property Not irradiated 10-2 Atom changes Vacancy Strain 10-6 Size (m) Diffusion and Interstitials growth Voids process 10-10 Atomic Collisions 10-14-Nuclear reactions 100nm 10-18 10-20 10-10 10-5 100 10-15 105 1010 Time Scale (sec)

### Multi-Scale Modeling

# Data flow between each code

Nuclear reactions (PHITS code)



Atomic collisions (Molecular dynamics)

Point defect distribution

Damage structural evolutions (Reaction kinetic analysis)



Concentration of defect clusters



Mechanical property change (Three-dimensional discrete dislocation dynamics)



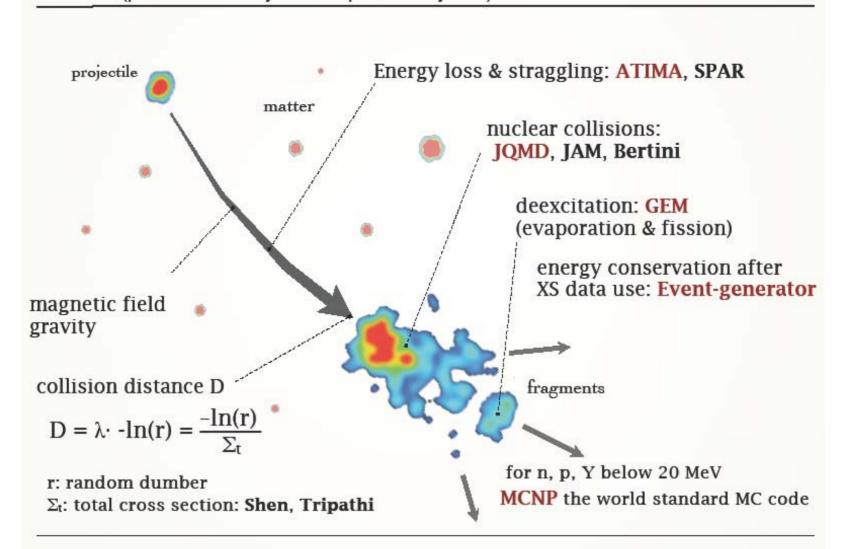
#### **Multi-Scale Modeling**

1. Nuclear Reaction

Nucleation rate of neutrons, photons, charged particles and PKA energy spectrum by them

PHITS (particle and heavy ion transport code system)

(4)



# Multi-Scale Modeling (5)

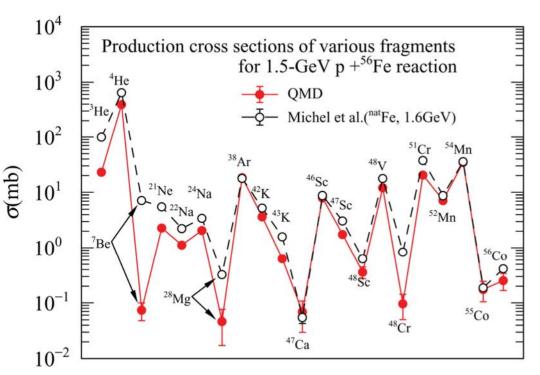
#### PHITS (Particle and Heavy Ion Transport code System) Feature (1): JQMD-1

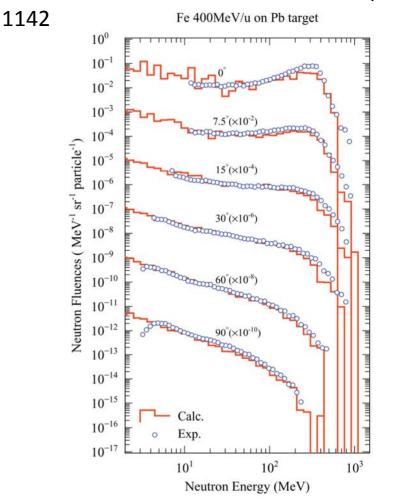
JQMD (Jaeri Quantum Molecular Dynamics) for Simulation of Nucleus-Nucleus Collisions
K. Niiita et.al. Phys. Rev. C52 (1995) 2620 http://hadron31.tokai.jaeri.go.jp/jgmd/

H. Iwase et.al. J. Nucl. Sci. Technol. 39 (2002)

Analysis of Nucleon-Nucleus Collisions by **JQMD** 

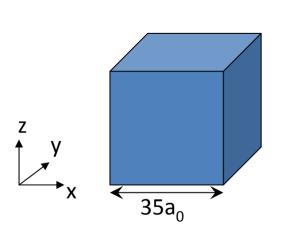
S. Chiba et.al. Phys. Rev. C53 (1996) 1824, C54 (1996) 285





## 2 Atomic Collision Simulation by MD

#### Calculation Condition



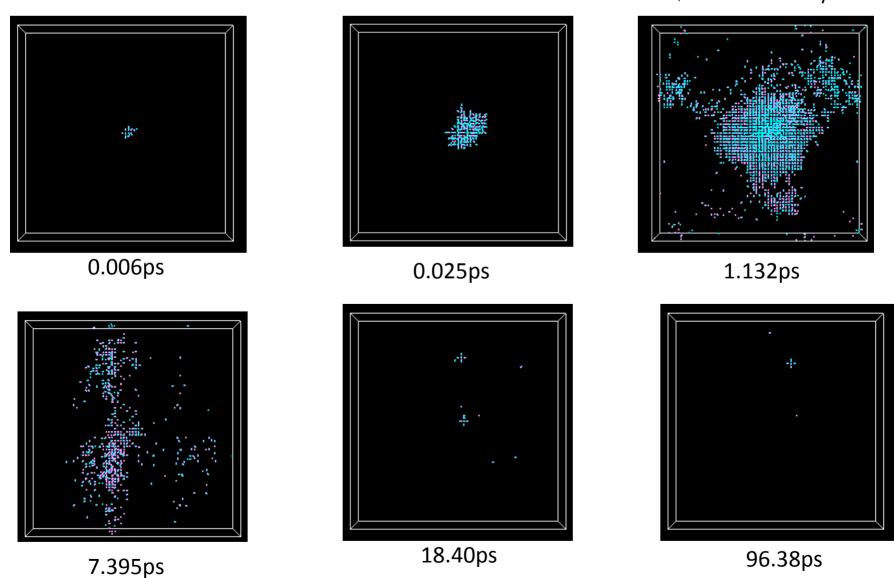
a<sub>0</sub>: lattice constant at 300 K

- > Potential model by Daw and Baskes (1984)
- NVE ensemble (i.e., number of the atoms, cell volume and energy were kept constant)
- > 35x35x35 lattices (171,500 atoms) in a simulation cell
- Periodic boundary condition for the three directions
- Initial condition : equilibrium for 50 ps, at 300 K, 0 MPa
- > PKA energy: 10keV
- MD runs with different initial directions of PKA (none of which were parallel to the lattice vector.)

#### **Multi-Scale Modeling**

# Typical Distribution of Point Defects

Marble: interstitial atoms, Violet: Vacancy cites



#### Multi-Scale Modeling (8)

#### 3. Damage Structure Evolution by Reaction Kinetic Analysis

$$\frac{dC_{I}}{dt} = P_{I} - 2Z_{I,I}M_{I}C_{I}^{2} - Z_{I,V}(M_{I} + M_{V})C_{I}C_{V}$$

damage production I-I recombination mutual annihilation

$$-Z_{I,IC}M_{I}C_{I}S_{I}-Z_{I,VC}M_{I}C_{I}S_{V}-M_{I}C_{I}C_{S}......$$

absorption by loops absorption by voids

#### annihilation of interstitials at sink

 $C_i$ : Interstitial concentration (fractional unit).

 $C_{V}$ : Vacancy concentration

Z: Cross section of reaction

M: Mobility

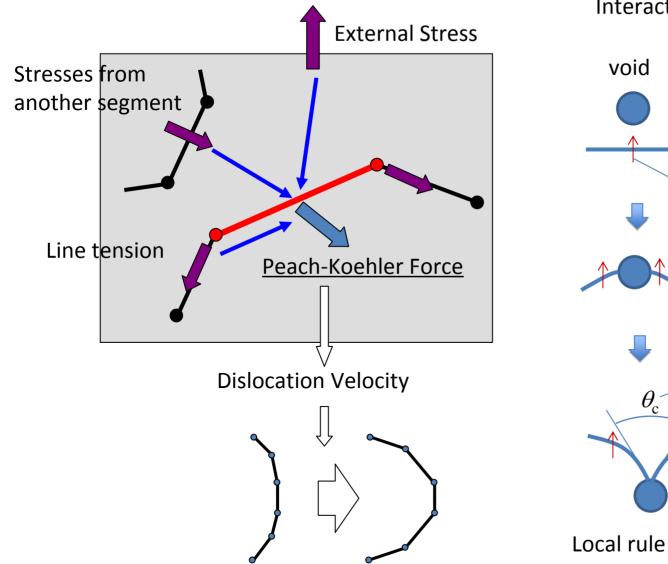
#### Multi-Scale Modeling (9)

# 4. Estimation of mechanical property changes with the 3D-Discrete Dislocation Dynamics code

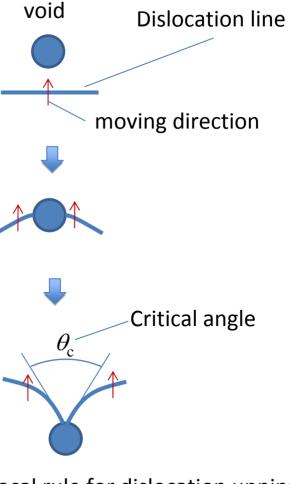
- Strengths of multilayered structures have been investigated using three-dimensional discrete dislocation dynamics (DDD) simulation.
- The multilayered structure was modeled as a stack of misfit dislocation networks which must exist at an interface between adjoining crystals having different lattice constants.
- Passages of a single mobile dislocation through several kinds of network stacks were simulated.
- The critical stress required for the dislocation passage depended on the dislocation spacing of the network, the number of network sheet and the spacing between network sheets.

#### Multi-Scale Modeling (10)

#### 3D-Discrete Dislocation Dynamics Procedure



Interaction with a void

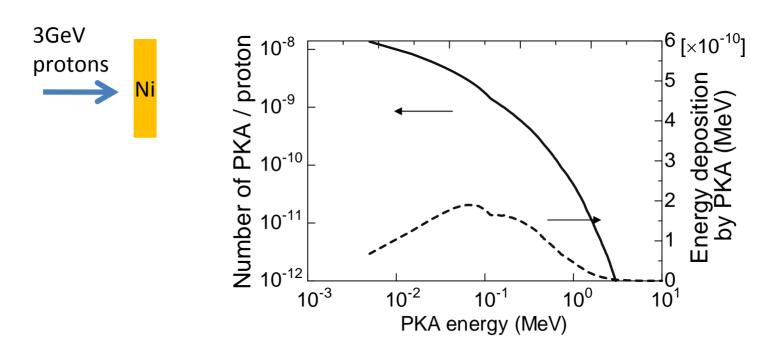


Local rule for dislocation unpinning

#### Multi-Scale Modeling (11)

### Sample Calculation in Case of Ni

#### **Result of PHITS Simulation**

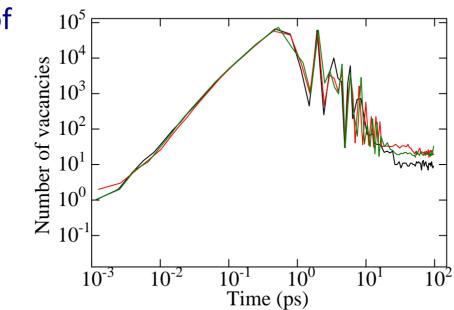


Number of PKA energy (left) and energy deposition by PKA (right) in 3 GeV proton irradiated Ni of 3 mm in thickness.

#### **Multi-Scale Modeling (12)**

#### Results of MD Calculation

Number of vacancies



- ➤ On average 17 vacancies, 17 interstitials were produced.
- Formation of defect clusters is calculated by k-Monte Carlo. Clusters of three point defects are formed.

# Results of k-Monte Carlo under the condition at 423K and 10 dpa:

Formation of vacancy clusters of four vacancies, concentration: 0.59x10<sup>-3</sup>,

Dislocation density: 1.1x10<sup>-9</sup>cm/cm<sup>3</sup>.

#### **Multi-Scale Modeling (13)**

#### Dislocation motion in the crystal with randomlydistributed voids

#### Normal stress

#### **OMPa**

200MPa

300MPa

400MPa

500MPa

600MPa

650MPa

700MPa

750MPa

800MPa

820MPa

840MPa

860MPa

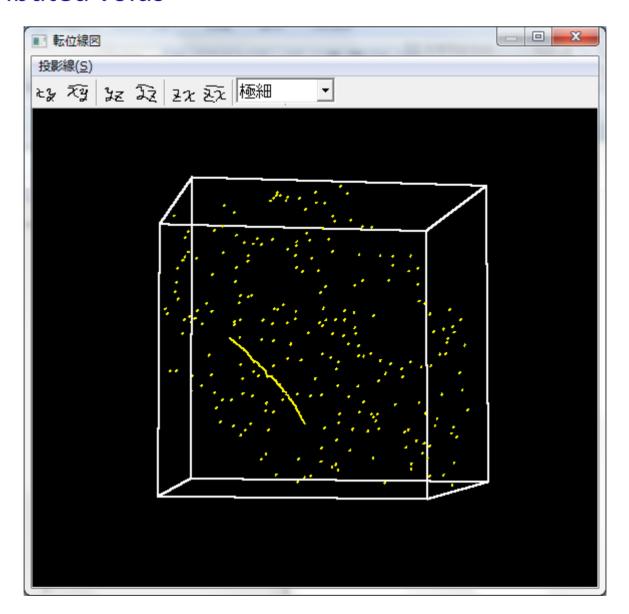
880MPa

900MPa

920MPa

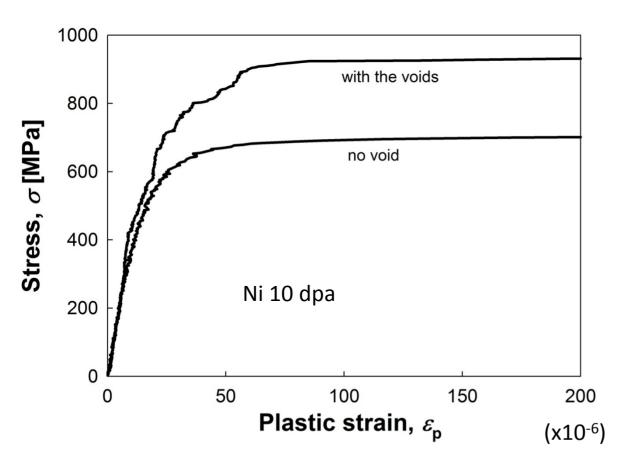
930MPa

935MPa



# Multi-Scale Modeling (14)

#### Stress-Strain Curves Calculated with DDD Simulation



Plastic shear strain

$$\gamma_{\rm p}$$
 = A b / V

Plastic strain

$$\varepsilon_{\rm p} = \gamma_{\rm p} \, S_{\rm f}$$

A: area swept by a dislocation

b: Burgers vector

V: volume of model crystal

S<sub>f</sub>: Schmid factor

Plastic strain is recalculated such that dislocation density  $\rho$ =1.1x10<sup>-9</sup>cm<sup>-2</sup>

Radiation hardening is expressed.

### **Summary**

# (1) Development of materials resistant to beam impact and radiation damage

- Successful to develop a remarkably strong nano-structured tungsten with RT ductility by means of MA, HIP and TMT for strengthening weak GBs in the recrystallized state.
- Successful to develop a stainless steel resistant to intergranular corrosion and having a possibility to be strong to radiation damage and high creep resistance, by means of GBE method.
- 316L GBEM was highly resistant to Pb-Bi erosion.

#### (2) Multi-scale model simulation system has been constructed.

- Nuclear reactions : PHITS, PKA energy spectrum
- Atomic collision : MD, k-Monte Carlo, Subcascade analysis.
   The number of point defects and clusters in the subcascade
- Damage structure evolution : Reaction kinetic analysis
   Vacancy clusters, Dislocation density
- Mechanical properties: Discrete dislocation dynamics
   Statistic energy calculation. Stress strain curve
- One-through calculation in case of nickel (typical fcc crystal)
- In future: Improve of parameters and DDD model considering interactions.

# Thank you for your attention!

#### **Acknowledgement:**

The present work has been performed under the FY2007 to FY2011 JSPS Scientific Research Grants categorized to "S" project, No. 19106017 entitled with "Comprehensive study on material damage mechanism by experimental and theoretical methods and development of materials for high-energy quantumbeam fields".

